

# Applications of Powder Diffraction Methods

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- ◆ Scattering sources, instruments, line shapes, resolution
- ◆ Rietveld Structure Refinement Analysis
- ◆ Magnetic Structures and Diffraction
- ◆ Example Refinement  $\text{Ho}_6\text{Fe}_{23}\text{D}_x$

# What do we need to do neutron scattering?

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- **Neutron Source – produces neutrons**
- **Diffractometer or Spectrometer**
  - Allows neutrons to interact with sample
  - Sorts out discrete wavelengths by monochromator (reactor) or by time of flight (pulse source)
  - Detectors pick up neutrons scattered from sample
- **Analysis methods to determine material properties**
- **Brain power to interpret results**

# Powder Diffraction

## Experimental View

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- ◆ Two types of diffractometers
  - Fixed Wavelength Diffraction (x rays, neutrons at a Reactor [HFIR, NCNR])
  - Variable wavelength (white beam neutrons) [beam may be pulsed (LANSCE [20 Hz], ISIS [50 Hz], SNS [60 Hz]) or continuous (PNS, Geneva)]
- ◆ Objective (determine “d” or “2θ”)  
Bragg Law  $\lambda = 2d \sin\theta$ 
  - $d = \lambda / (2 \sin \theta)$ 
    - » pulse neutron source - vary  $\lambda$ , keep  $\theta$  fixed
    - » x-ray or reactor neutron source -- vary  $\theta$ , keep  $\lambda$  fixed

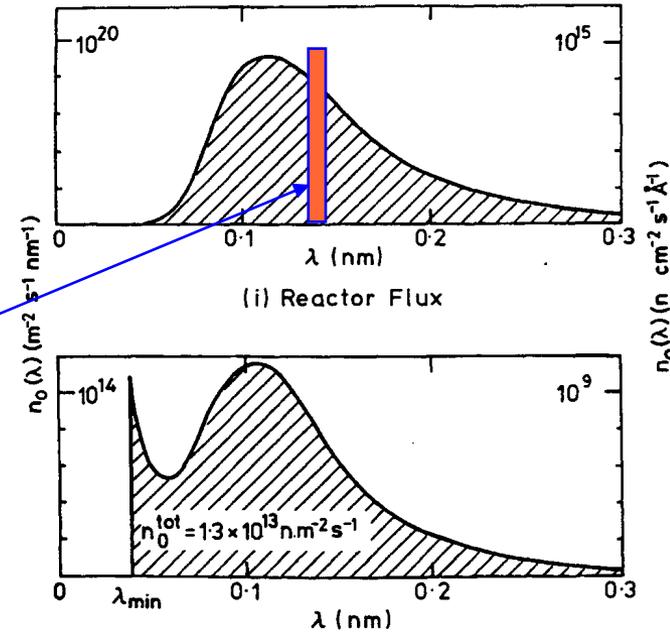
# Neutron sources – steady state (Reactors) and pulsed (Spallation)

## ◆ Reactor

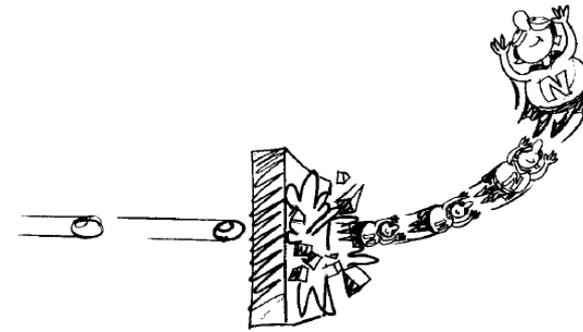
- Fission of  $U^{235}$  produces neutrons
- Fission spectrum moderated (slowed down) by either  $D_2O$  or  $H_2O$  (less good moderator) and neutrons are extracted through beam tubes for spectrometers – fixed wavelength used

## ◆ Spallation source

- High E protons (e.g., 800 MeV) impinge on target (W, Hg or U)
- Nucleus of target is raised to excited state and subsequent decay produces neutrons (+  $\gamma$ s, nucleons and neutrinos) – 15 – 25 neutrons produced per proton with average E = 55 MeV
- Neutrons moderated by liquid H,  $H_2O$  or methane
- Spallation sources generally operate in pulse mode – 20 Hz at LANSCE, 60 Hz at new SNS



Time of flight is used to sort out wavelengths

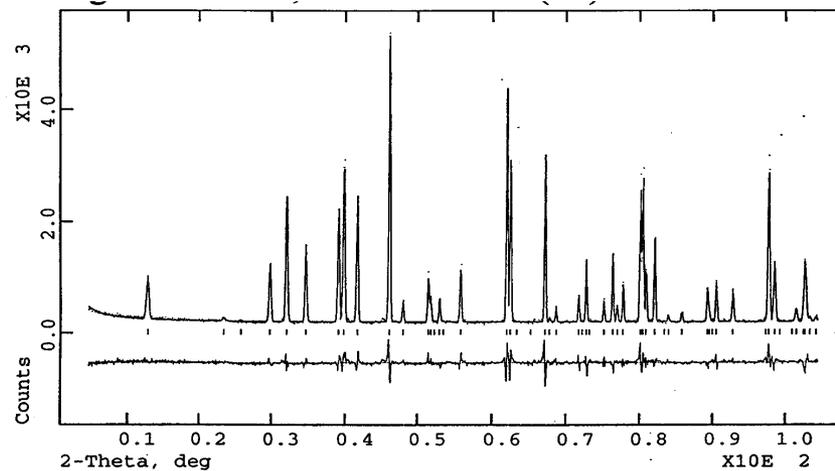
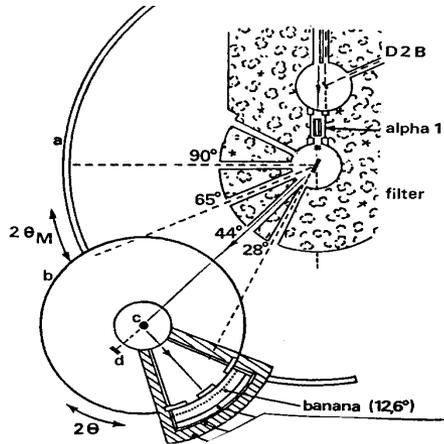


# COMPARISON OF POWDER DIFFRACTION INSTRUMENTS

\*\*\* Bragg Scattering:  $\lambda = 2d \sin \theta$  [we need "d"] \*\*\*

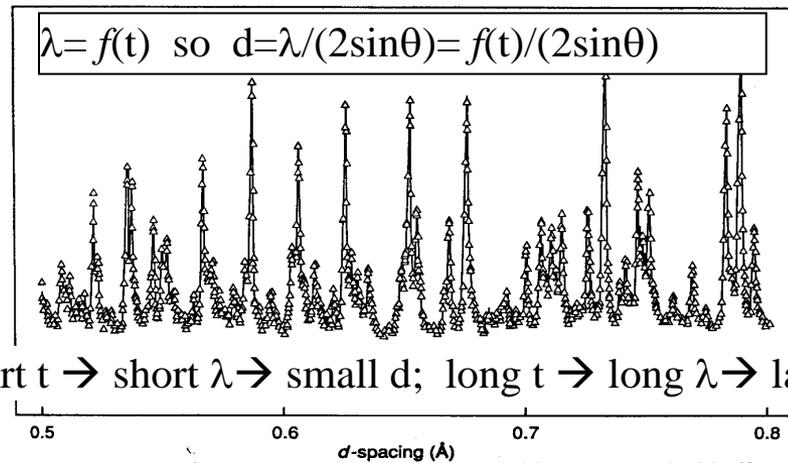
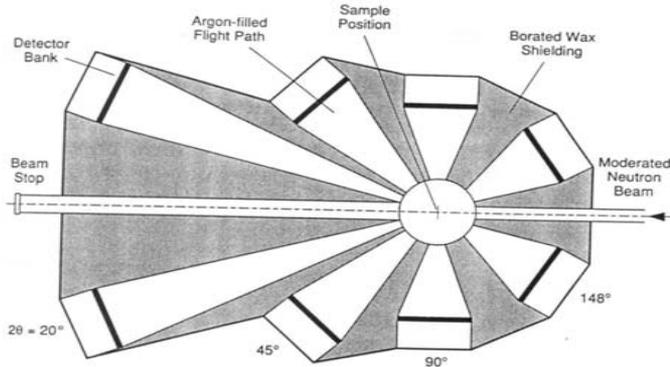
Reactor Source -- Monochromator selects near mono-energetic neutrons, detector moves ( $2\theta$ ) to collect discrete diffraction data

• ( $\lambda$  fixed, vary  $\theta$ )



• Pulse Source -- White beam of moderated neutrons used, neutron time-of-flight selects wavelength, detectors grouped in banks, no moving parts

( $\theta$  fixed, vary  $\lambda$ )



short  $t \rightarrow$  short  $\lambda \rightarrow$  small  $d$ ; long  $t \rightarrow$  long  $\lambda \rightarrow$  large  $d$

# Using Powder Diffraction

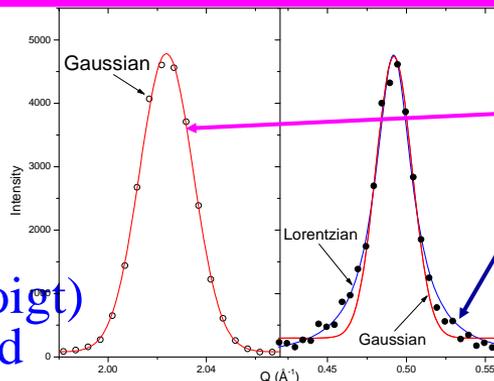
## Input Information -- Structure Determination

### ◆ Know instrument-dependent scattering line-shape

- Gaussian for  $\lambda$  fixed
- Sample distortions (pseudo Voigt) linear comb. of Lorentzian and Gaussian
- Convolution of rising and falling exponentials with Gaussian for TOF

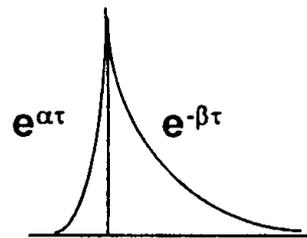
### ◆ Dependence of resolution on “d” [or Q] varies widely with instrument design

- For pulse sources  $\Delta d/d$  near constant
- For reactor – resolution has minimum



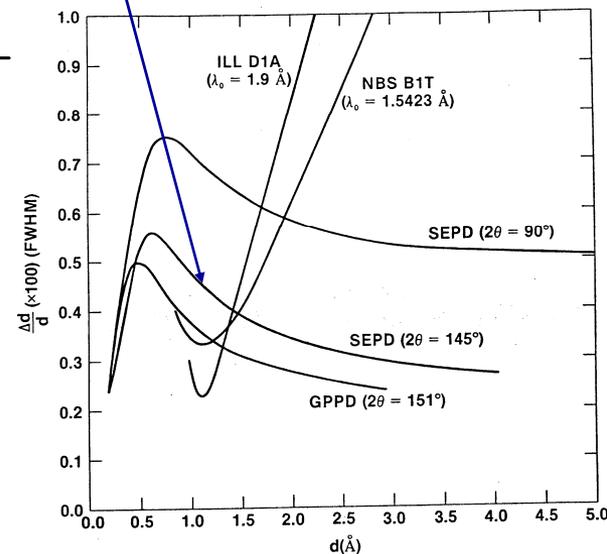
$$R(Q) = \sqrt{\frac{4 \ln 2}{\pi \Gamma^2}} e^{-\left(\frac{4 \ln 2 (Q - Q_0)^2}{\Gamma^2}\right)}$$

$$R(Q) = \frac{2}{\pi \Gamma'} \frac{1}{1 + \left(\frac{2(Q - Q_0)}{\Gamma'}\right)^2}$$



$$R(T) = N \left[ e^u \operatorname{erfc}(y) + e^v \operatorname{erfc}(z) \right]$$

(VonDreele, Jorgensen & Windsor)



# Rietveld Profile Refinement

## Least Squares Fitting Procedure for Powder Data

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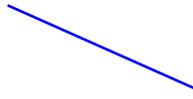
### ◆ Input Data

- Powder scattering pattern data
- Trial structure space group and approximate lattice parameters and atomic positions [coordinates of atoms in cell - may be variables x,y,z]
- Line shape function and Q-dependence of resolution

### ◆ Output Results

- Lattice Parameters
- Refined atomic positions and occupancies
- Thermal parameters (Debye Waller) for each atom site
- Line shape parameters (may reflect sample strains, etc.)
- Background parameters
- R factors of fit (and other measures of fit precision)
- Preferential orientation, absorption, etc.

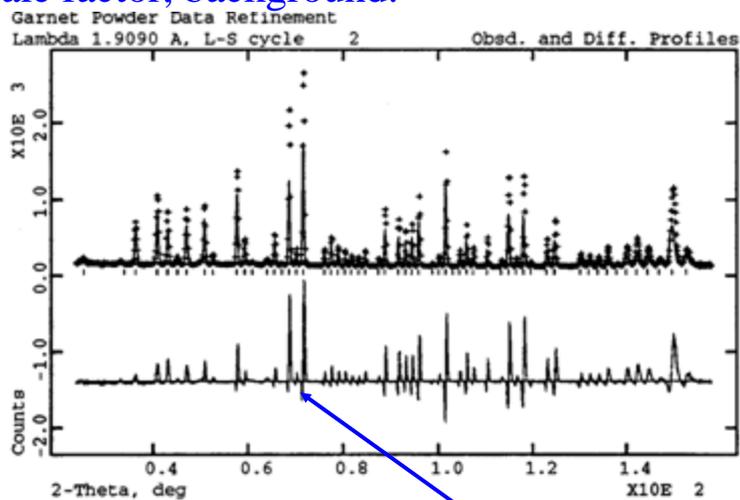
e.g., Caglioti, Paoletti & Rice


$$\Gamma^2 = U \tan^2 \theta + V \tan \theta + W$$

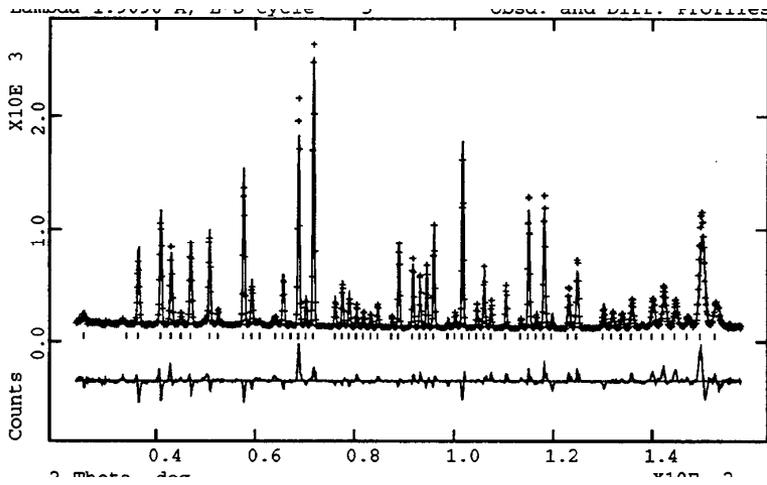
### ◆ More than one phase can be separately refined

# Progress of a Refinement

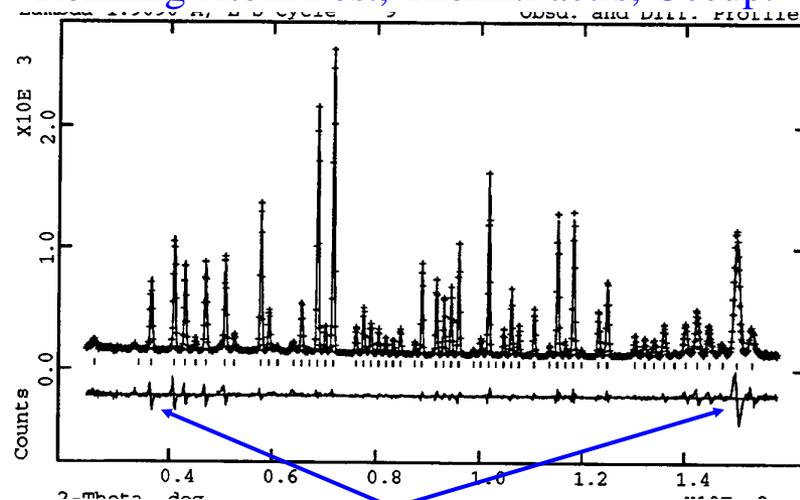
First Cut -- Input data and basic info and refine scale factor, background.



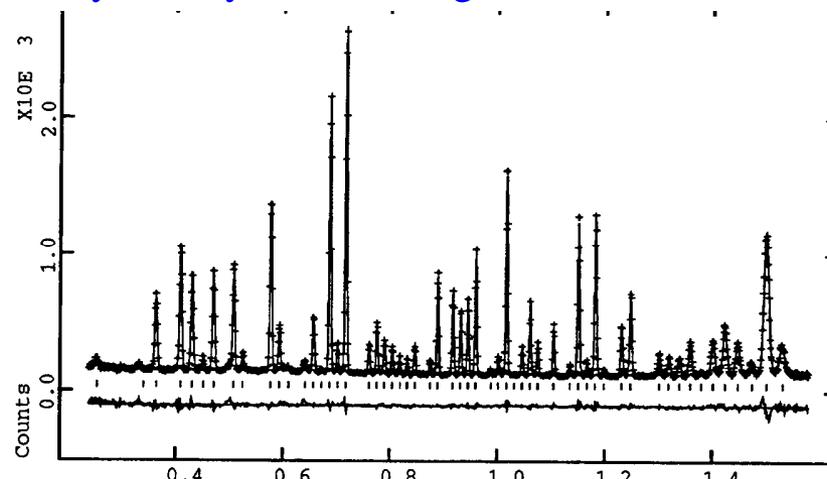
Next Refine Lattice Parameters and Zero Point



Refining Atom Pos., Therm.Factors, Occup.



Refine Resolution Parameters, Peak Asymmetry, More background coefficients



# Cautions with Rietveld

## (i.e. don't blindly believe the results)

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### ◆ Hazards

- Incorrect space group -- may only slightly degrade R factors, symptoms -- unrealistically large thermal factors, misfit on some peaks not others
- Parameter correlation and compensation -- e.g., absorption and thermal parameters may correlate -- look at variance-covariance matrix (if available from program)
- Refining multiple phases may be unrealistic -- get a better sample!
- Over parameterization -- Simultaneous refining of too many parameters may lead to lack of uniqueness

# Profile Fitting Programs

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- ◆ Original Description of Method -- H.M. Rietveld, J. Appl. Cryst. **2**, 65 (1969)
- ◆ *FULLPROF* [good for magnetic structures] -- Juan Rodriguez-Carvajal, ILL
- ◆ Defacto World Standard -- *Generalized Structure Analysis System [GSAS]* -- A.C. Larson & R.B. Von Dreele, LANL and ANL

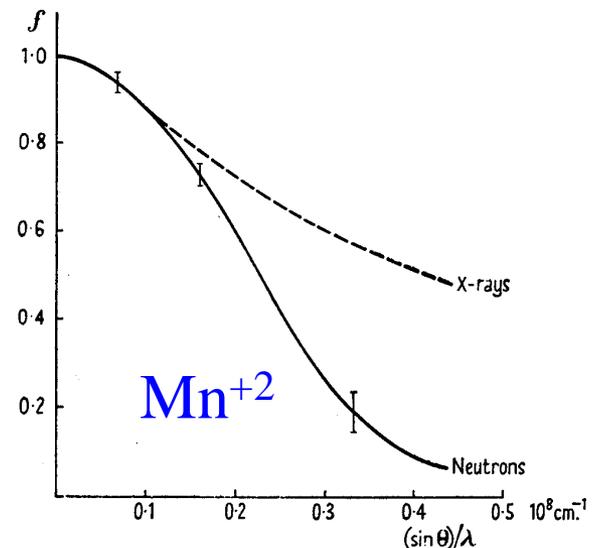
# Magnetic Powder Diffraction

- ◆ Neutron has a magnetic moment -- will interact with any magnetic fields within a solid, e.g., exchange field
- ◆ Equivalent (to b) magnetic scattering amplitude for an atom

$$p = \left( \frac{e^2 \gamma}{2mc^2} \right) fgJ = 0.269 \times 10^{-12} fgJ (\text{cm})$$

where  $g$  = Lande “g” factor,  $J$  = total spin angular momentum,  $f$  = magnetic electrons form factor

- ◆ Magnetic scattering comes from polarized spins (e.g., 3d [Fe] or 4f [RE]) not from nucleus -- Therefore scattering amplitude is Q-dependent (like for x-rays) via  $f$
- ◆ at  $Q = 0$  for Fe  $\mu = gJ = 2.2$  Bohr magneton  
 $p = 0.6$  (comparable to nuclear  $b = 0.954$ )  
 all in units of  $10^{-12}\text{cm}$

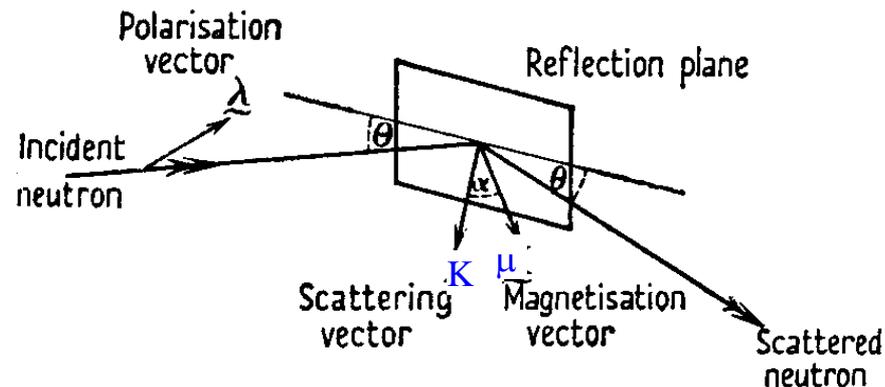


# Magnetic Powder Diffraction II

- ◆ In diffraction with unpolarized neutrons (polarized scattering is a separate topic) the nuclear and magnetic cross sections are independent and additive:

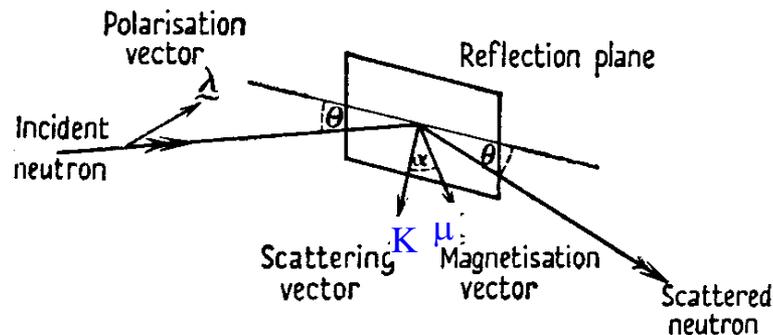
$$\frac{d\sigma}{d\Omega} \propto S(Q) = S_{nucl} + q^2 S_{magn} = S_{nucl} + (1 - \cos^2 \alpha) S_{magn}$$

- ◆  $q^2$  is a “switch” reflecting fact that only the component of the magnetic moment  $\mu \perp$  scattering vector  $\mathbf{K}$  (or  $\mathbf{Q}$ ) contributes to the scattering



# $q^2$ “switch” affects intensity on Bragg peaks in a polycrystal (powder)

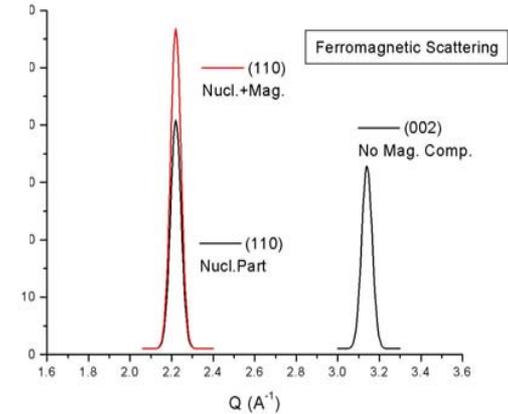
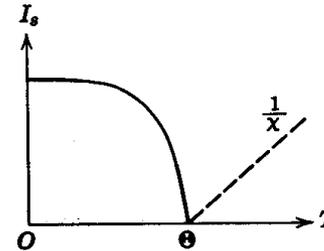
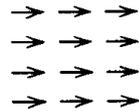
- ◆ **Polycrystalline averaging** over  $q^2 = (1 - \cos^2\alpha)$  for various symmetries
  - Cubic:  $q^2 = 2/3$  [no moment direction information]
  - Tetragonal:  $q^2 = 1 - d^2[1/2(h^2+k^2)a^{*2}\sin^2\varphi+l^2c^{*2}\cos^2\varphi]$   
 $\varphi =$  angle between  $\mu$  and  $c$  axis
  - Hexagonal:  $q^2 = 1 - d^2[1/2(h^2+k^2+hk)a^{*2}\sin^2\varphi+l^2c^{*2}\cos^2\varphi]$
  - ...



# How is magnetism reflected in a scattering pattern?

## • Long-range ordered

- Ferromagnet (Fe, Ni, Gd)
- Spins parallel on lattice



## – Ferrimagnet ( $TbFe_2$ , $Fe_3O_4$ )

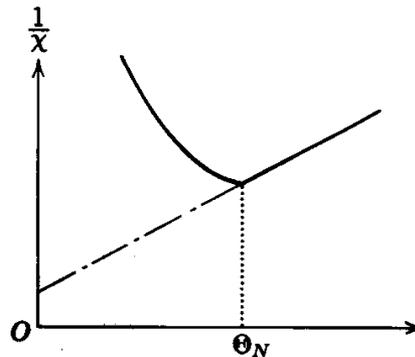
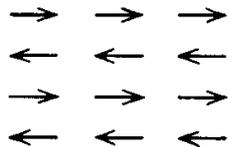


- Spins parallel on each site
- Sites can reverse spin direction
- $\Sigma\mu \neq 0$  on each site, but overall  $\Sigma\mu$  could = 0

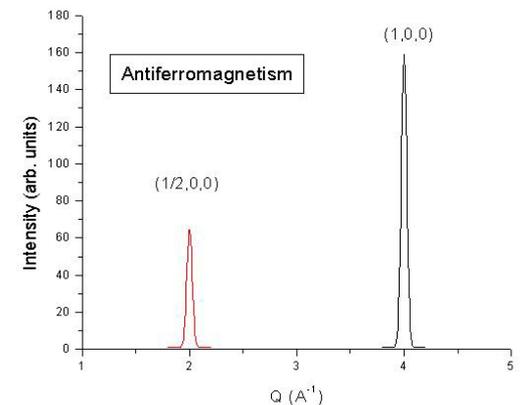
Scattering only at Bragg peak positions (adds to nuclear), but not necessarily all ( $q^2$  switch)

## • Antiferromagnet ( $MnO$ , $MnF_2$ )

(parallel spins with alternate sites reversed in direction)



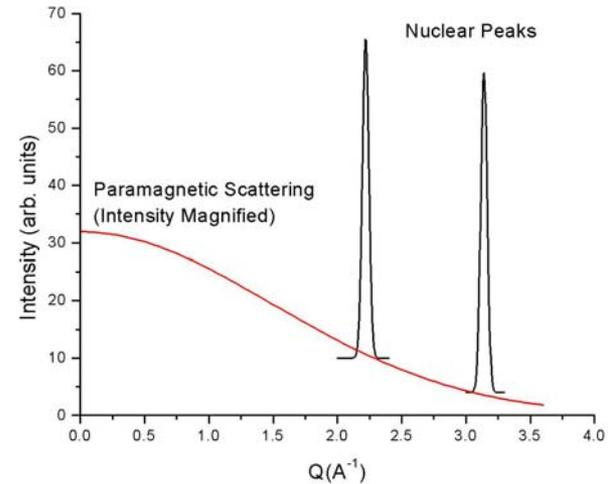
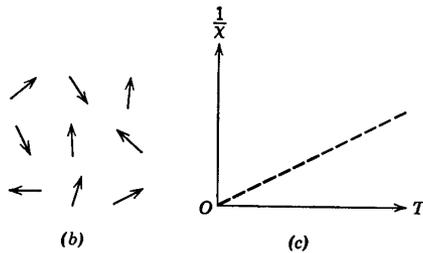
Net moment = 0  
Job security for neutron scatterers



- Magnetic unit cell doubled wrt crystal cell
- Purely magnetic scattering peaks at half Miller index positions

# How is magnetism reflected in a powder pattern?

- **Paramagnetism ( $T > T_C$ )**
  - spins thermally disordered



- no coherent magnetic scattering
- Incoherent neutron scattering (weak) has form factor dependence

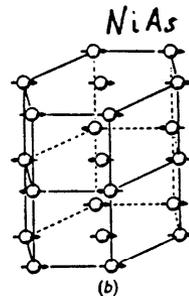
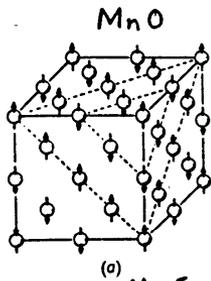
# We can make life more complicated with magnetic structures

- ◆ Complex antiferromagnetic cells (multi-site, but on each site  $\Sigma\mu = 0$ )

- ◆ Periodic Moment Structures (e.g., Er,  $\text{Au}_2\text{Mn}$ )

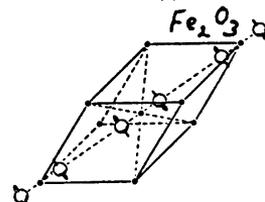
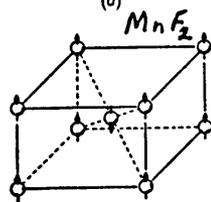
- $\Sigma\mu = 0$ , but periodic spin arrangement gives rise to magnetic scattering satellites (+, -) of nuclear Bragg peaks

(111) FM Sheets

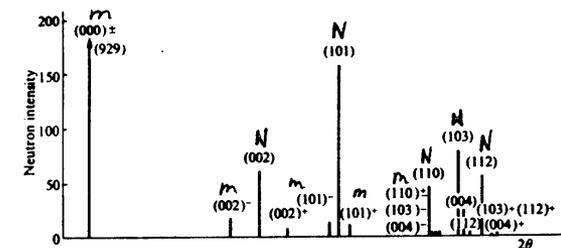
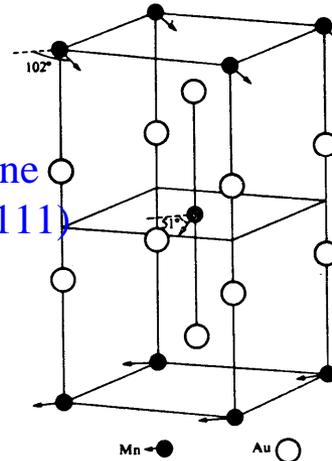


FM layers along (0001)

bcc site antiparallel to corners



AFM line along (111)



# Magnetic structure refinement

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- ◆ Nuclear and magnetic intensity are additive

$$\frac{d\sigma}{d\Omega} \propto S(Q) = S_{nucl} + q^2 S_{magn} = S_{nucl} + (1 - \cos^2 \alpha) S_{magn}$$

- ◆ Magnetic structure can be refined using GSAS as a separate “phase” alongside the atomic structure
- ◆ General magnetic structure factor depends on Shubnikov-Dehaas groups

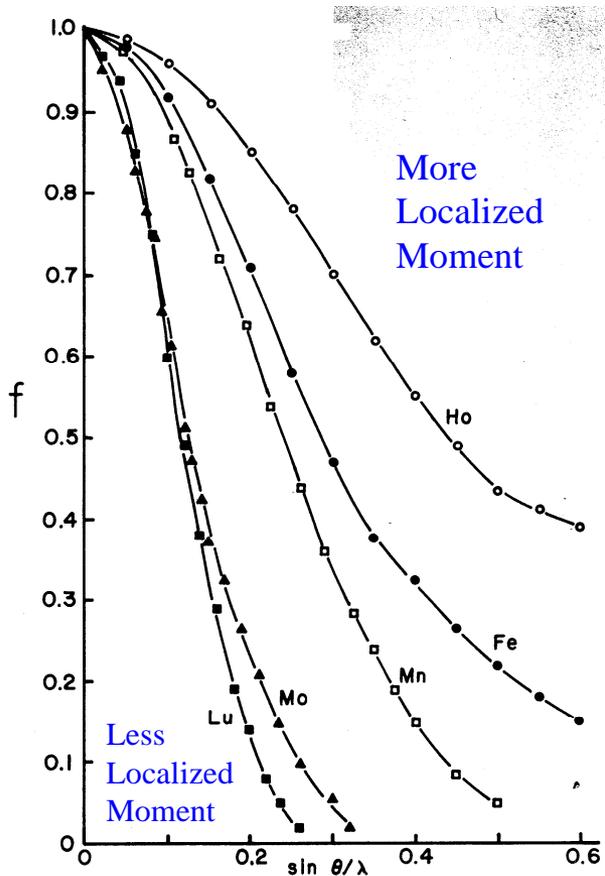
- Magnetic scattering amplitude depends on moment and form factor

$$p = \left( \frac{e^2 \gamma}{2mc^2} \right) fgJ = 0.269 \times 10^{-12} fgJ (cm)$$

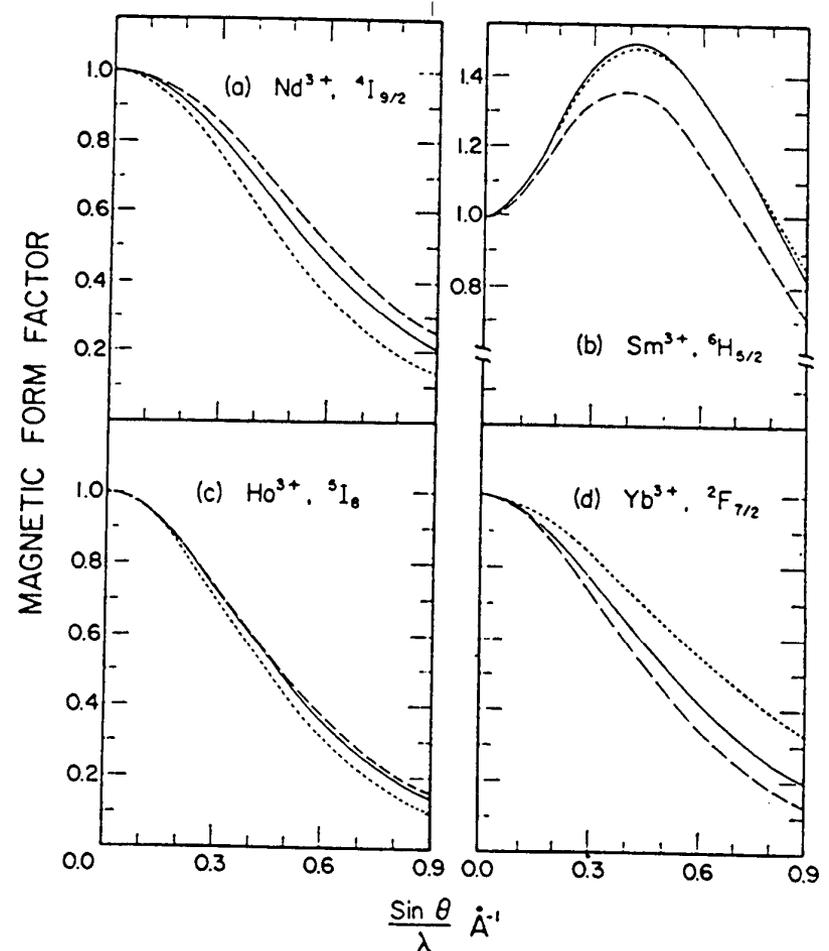
- Form factors,  $f$ , are determined experimentally for most magnets (e.g., RE's) -- can be calculated from electronic band structure and other techniques
- Refinement gives moment magnitudes on each site and x,y,z components (if symmetry permits)

# Form Factors

## ◆ Experimental



## ◆ Calculated



# Structure determination case study

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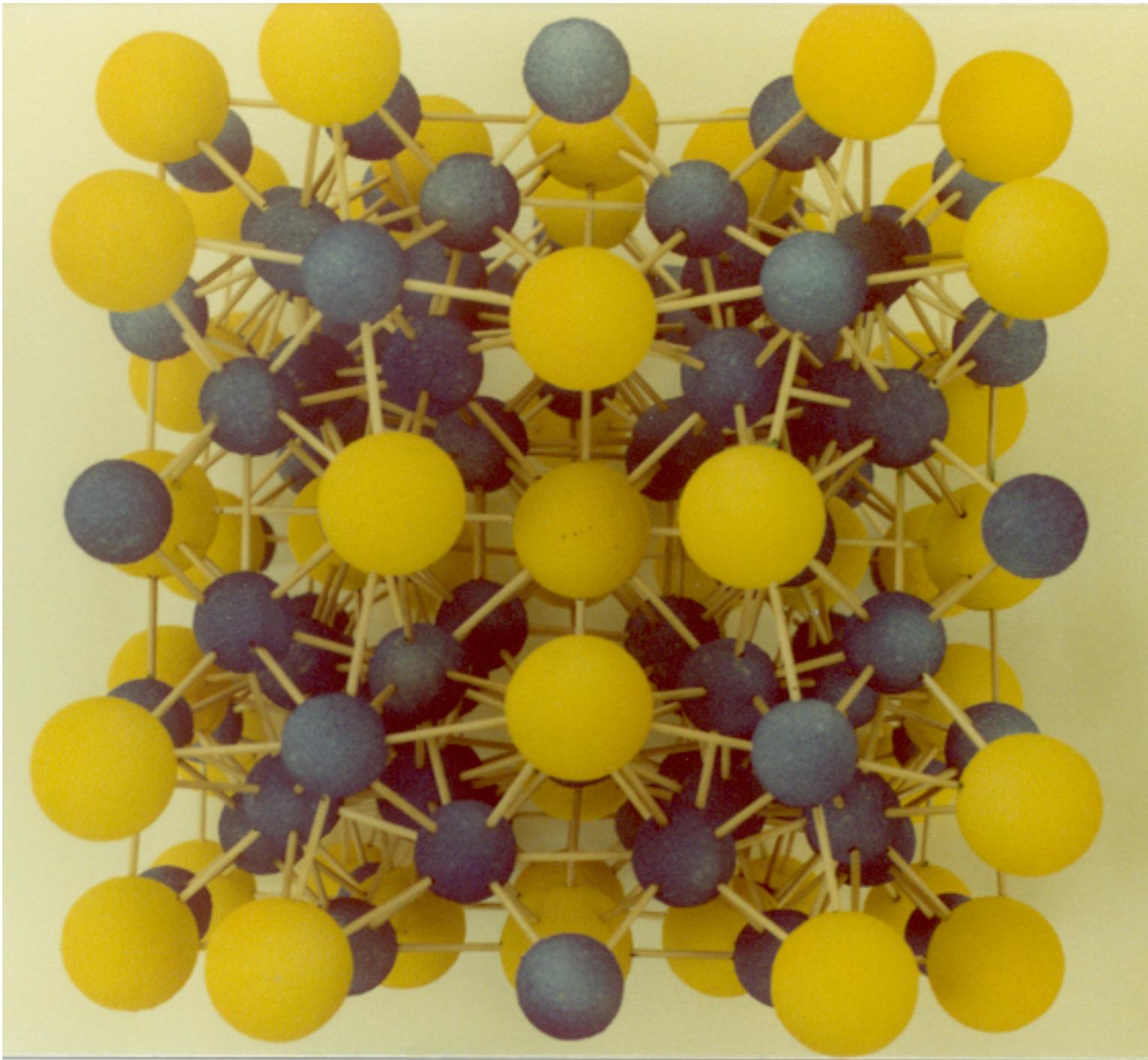
Rare earth intermetallic  
compounds

$\text{Ho}_6\text{Fe}_{23}$  and  $\text{Ho}_6\text{Fe}_{23}\text{D}_x$

# Structure Refinement Example



- ◆ Rare Earth Intermetallic Compound
- ◆ fcc cell - Fm3m structure (space group)
- ◆ 4 Formula Units per Unit Cell
- ◆ Large Unit Cell --  $a = 12 \text{ \AA}$
- ◆ Absorbs H (or D) from 1 up to 16 atoms/FU -- forms stable hydride compounds
- ◆ Also magnetic (both Ho and Fe)



# Ho<sub>6</sub>Fe<sub>23</sub> and Ho<sub>6</sub>Fe<sub>23</sub>D<sub>x</sub> (Fm3m) Unit Cell

4 F.U./cell

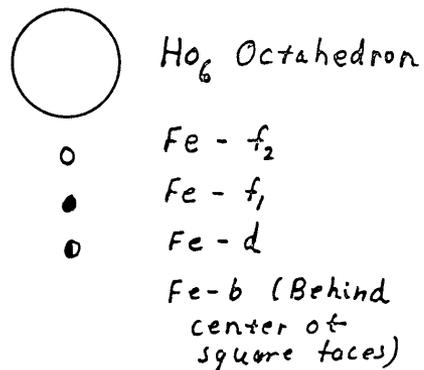
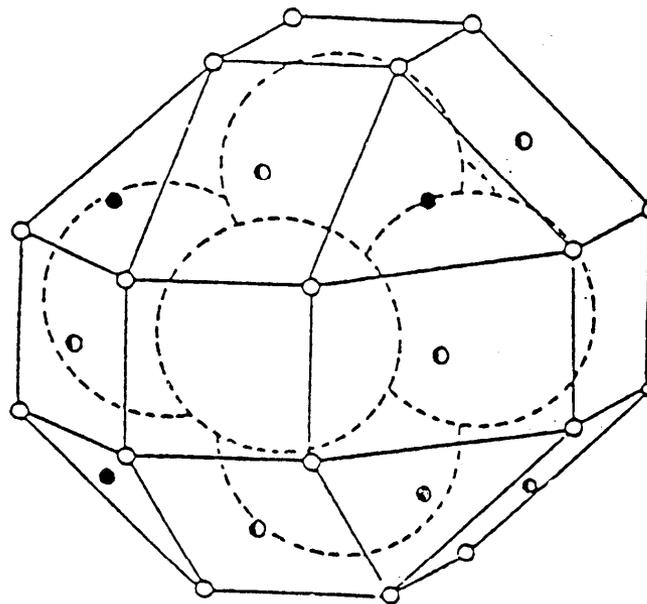
16 metal atoms (4 sites - all full)

324 possible D atoms (5 sites)

– only 64 D atoms in fact

-- some sites are empty or

-- some sites are partially occupied



FCC (Fm3m)		
ATOM	WYCOFF NOTATION	PARAMETERS
Ho	(24E)	(x, 0, 0)
Fe	(4B)	(.5, .5, .5)
Fe	(24D)	(0, .25, .25)
Fe	(32F <sub>1,2</sub> )	(x, x, x)



## Occupied Crystallographic Sites (13 allowed)

<i>F.c.c. (Fm3m)</i>		
<i>Atom</i>	<i>Wyckoff notation</i>	<i>Parameters</i>
Ho	24e	( <i>x</i> , 0, 0)
Fe	4b	(0.5, 0.5, 0.5)
Fe	24d	(0, 0.25, 0.25)
Fe	32f <sub>1,2</sub>	( <i>x</i> , <i>x</i> , <i>x</i> )
D	4a	(0, 0, 0)
D	32f <sub>3</sub>	( <i>x</i> , <i>x</i> , <i>x</i> )
D	96j <sub>1,2</sub>	(0, <i>y</i> , <i>z</i> ) <sub>1,2</sub>
D	96k	( <i>x</i> , <i>x</i> , <i>z</i> )

- Reitveld least squares refinement procedure determines
  - Lattice parameters
  - Atomic position coordinates
  - Site occupation numbers vs. maximum
  - Thermal (Debye-Waller) factors for each atom
  - Instrumental parameters (background, peak shape parameters)
- Note: Ho, Fe sites all full  $4x(6+23) = 116$
- D sites not full – 324 allowed > max  $4x16 = 64$

# Ho<sub>6</sub>Fe<sub>23</sub>D<sub>15.7</sub> Refinement Parameters I

Lattice Parameters, Volume Expansion  
relative to Ho<sub>6</sub>Fe<sub>23</sub>,

Ho<sub>6</sub>Fe<sub>23</sub>:

T(K)	a <sub>0</sub>	ΔV/V <sub>0</sub>	R <sub>e</sub>	R <sub>wp</sub>	χ
295	12.029	0	6.1	11.5	1.9
4.2	12.000	0	8.9	18.0	2.0

Ho<sub>6</sub>Fe<sub>23</sub>D<sub>15.7</sub>:

T(K)	a <sub>0</sub>	ΔV/V <sub>0</sub> (%)	R <sub>e</sub>	R <sub>wp</sub>	χ
295	12.423	9.8	5.1	9.2	1.8
4.2	12.406	10.2	3.1	9.8	3.2

R<sub>e</sub> = statistical variance for a perfect fit with taking into account the statistical precision of the data.

R<sub>wp</sub> = weighted profile variance of the fit (big pks. vs little pks.)

χ = ratio of R<sub>wp</sub>/R<sub>e</sub> (values near 2 are generally indicative of a good profile fit to the model structure)

# Ho<sub>6</sub>Fe<sub>23</sub>D<sub>15.7</sub> Refinement Parameters II

Refined atomic coordinates, thermal factors, and occupancy factors

(D) – Values are per formula unit – 4 formula units/unit cell

Atom	Wyckoff Site <b>Max. Occup.</b>	N	X(or Y)	Z	B
All sites full	Ho 24e <sub>(x,0,0)</sub>	6	0.205	(f)	--
	Fe 4b <sub>(0.5,0.5,0.5)</sub>	1	(f)	(f)	--
	Fe 24d <sub>(0,0.25,0.25)</sub>	6	(f)	(f)	--
	Fe 32f <sub>1(x,x,x)</sub>	8	0.177	=x	--
	Fe 32f <sub>2(x,x,x)</sub>	8	0.372	=x	--
	D 4a <sub>(0,0,0)</sub>	0	(f)	(f)	n/a
Sites part full	D 32f <sub>3(x,x,x)</sub>	7.2	0.096	=x	0.55
	D 96j <sub>1(0,y,z)</sub>	4.9	0.135	0.340	0.19
	D 96j <sub>2(0,y,z)</sub>	3.1	0.076	0.385	0.29
	D 96k <sub>(x,x,z)</sub>	1.4	0.159	0.047	0.94

# Ho<sub>6</sub>Fe<sub>23</sub>D<sub>15.7</sub> Magnetic Moments

- Refined magnetic structure confirms a colinear spin arrangement
- Ho moments (1 site – 24e) antiparallel to Fe moments (4 sites – 4b, 24d, 32f<sub>1</sub>, 32f<sub>2</sub>)
- Refinement produces site specific values of magnetic moment [Note: Total magnetic moment may have large errors due to compounding of errors from each site.]
- Moments at 4.2 K:

Atom	Wyckoff Site	N	Mom. ( $\mu_B$ )
Ho	24e	6	-10.2
Fe	4b	1	2.2
Fe	24d	6	2.4
Fe	32f <sub>1</sub>	8	2.0
Fe	32f <sub>2</sub>	8	2.2